

Short and Intermediate Range Structure at High Pressure: the QHP-PDF

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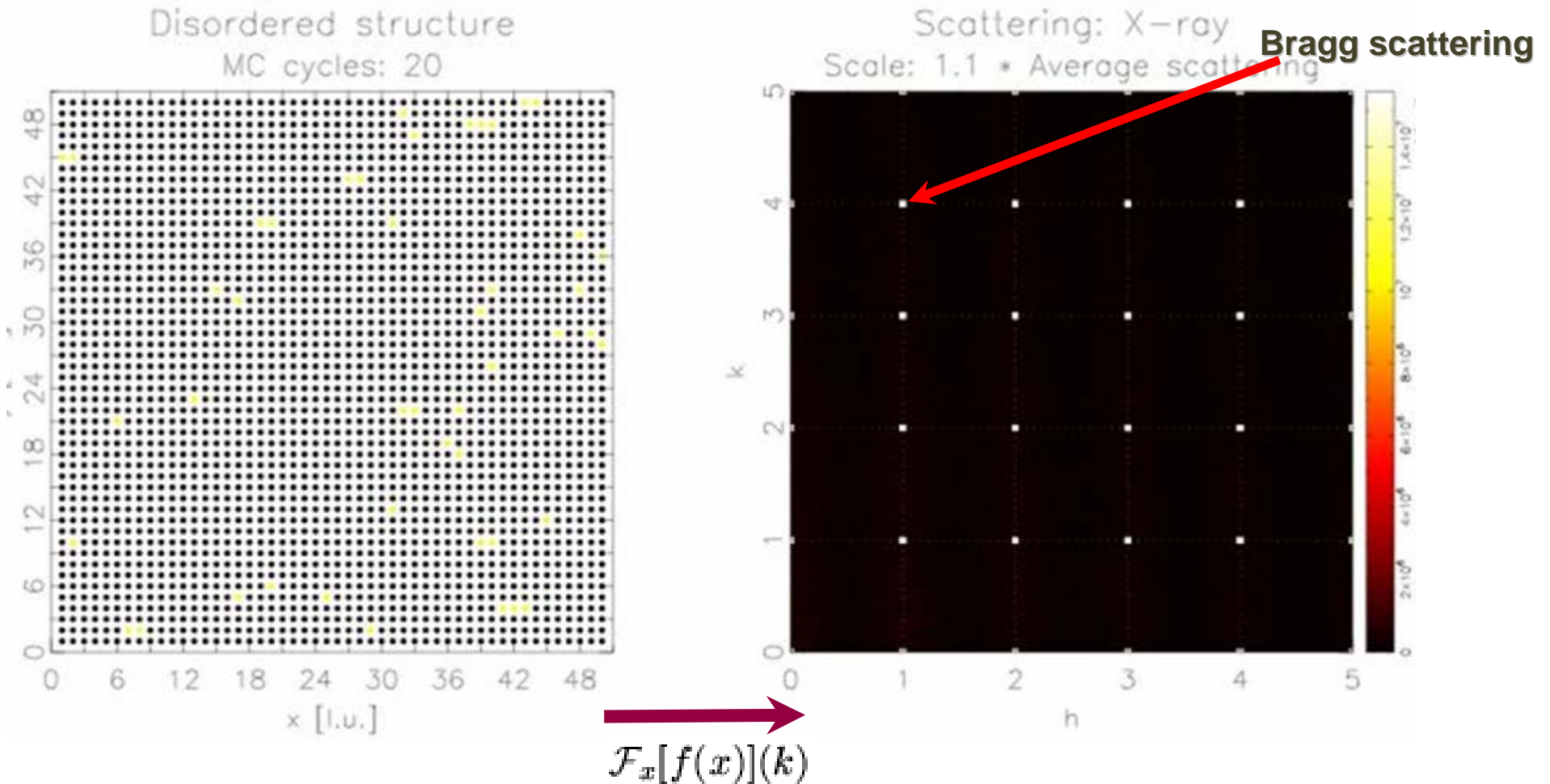
***One advantage of the pressure variable in studying
crystalline, nano-, glassy materials***

***- pressure varies short, intermediate and long range order
(usually) without changing chemistry***

Funding: NSF CHE-0221934 (CEMS), NSF-DMR-0452444,
DE-FG02-03ER46085

Perfect crystals (or random, uncorrelated disorder)

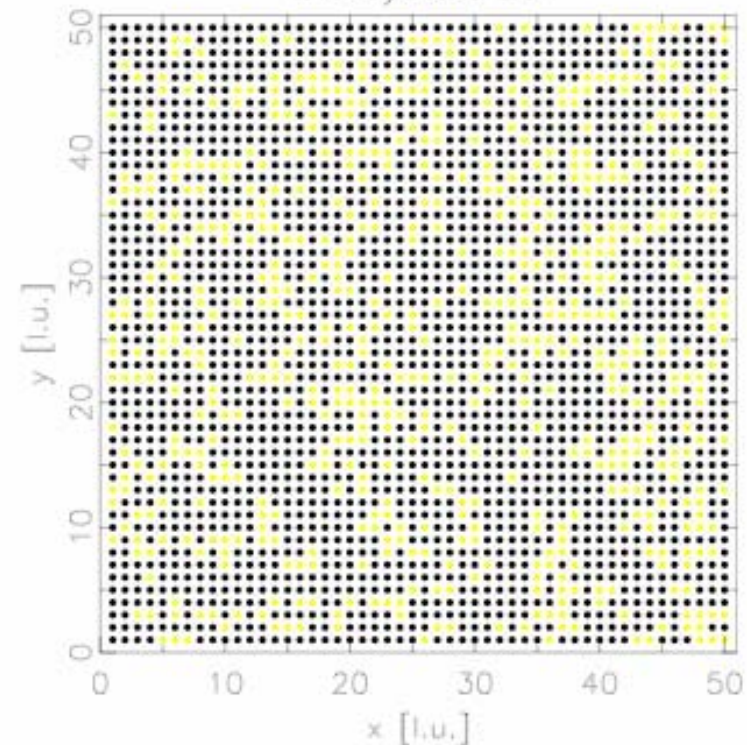
1% vacancies, No correlation only Bragg scattering



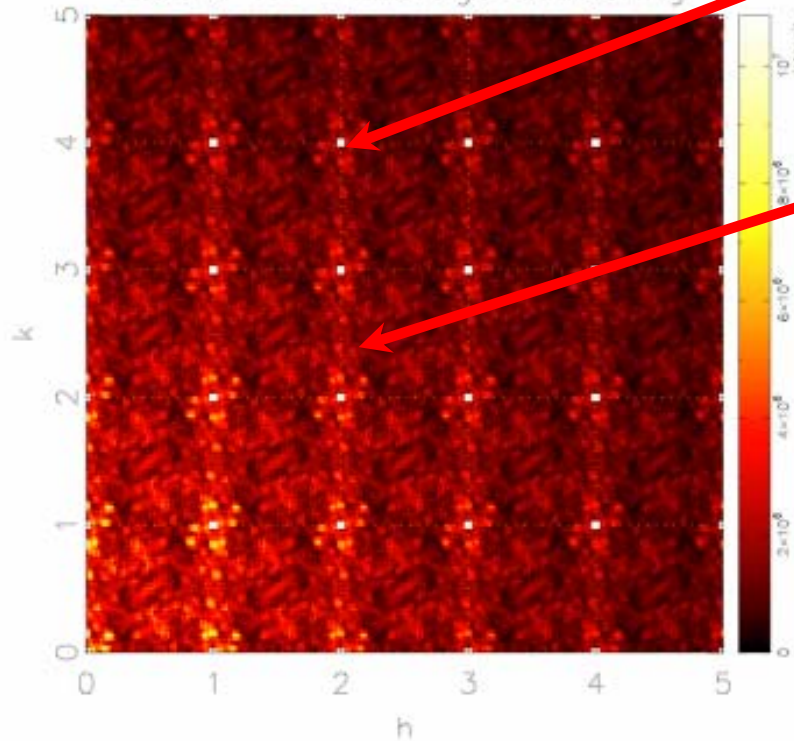
<http://www.totalscattering.org/teaching/>

24% vacancies: correlation unavoidable

Disordered structure
MC cycles: 20



Scattering: X-ray
Scale: 1.1 * Average scattering



Bragg scattering

Diffuse Scattering

Concentration achieved (%) : 24.040

CORRELATIONS :

$\alpha(100)$:	Target:	0.000	-	Achieved:	0.093
$\alpha(010)$:	Target:	0.000	-	Achieved:	0.025
$\alpha(110)$:	Target:	0.000	-	Achieved:	0.034
$\alpha(200)$:	Target:	0.000	-	Achieved:	0.104
$\alpha(020)$:	Target:	0.000	-	Achieved:	-0.025

BRAGG INTENSITIES

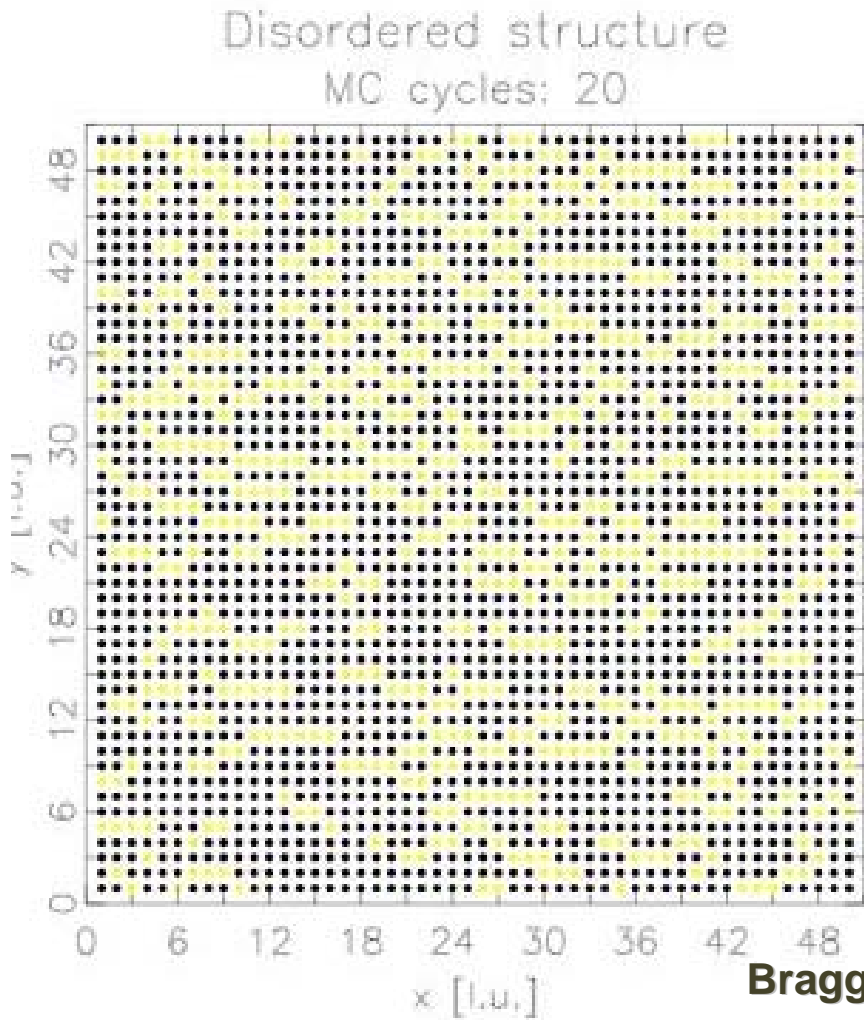
$(1\ 0\ 0)$:	0.72505897E+10
$(0\ 1\ 0)$:	0.72516050E+10
$(4\ 0\ 0)$:	0.36144604E+10
$(0\ 4\ 0)$:	0.36149660E+10

<http://www.totalscattering.org/teaching/>

Random and correlated disorder in scattering space

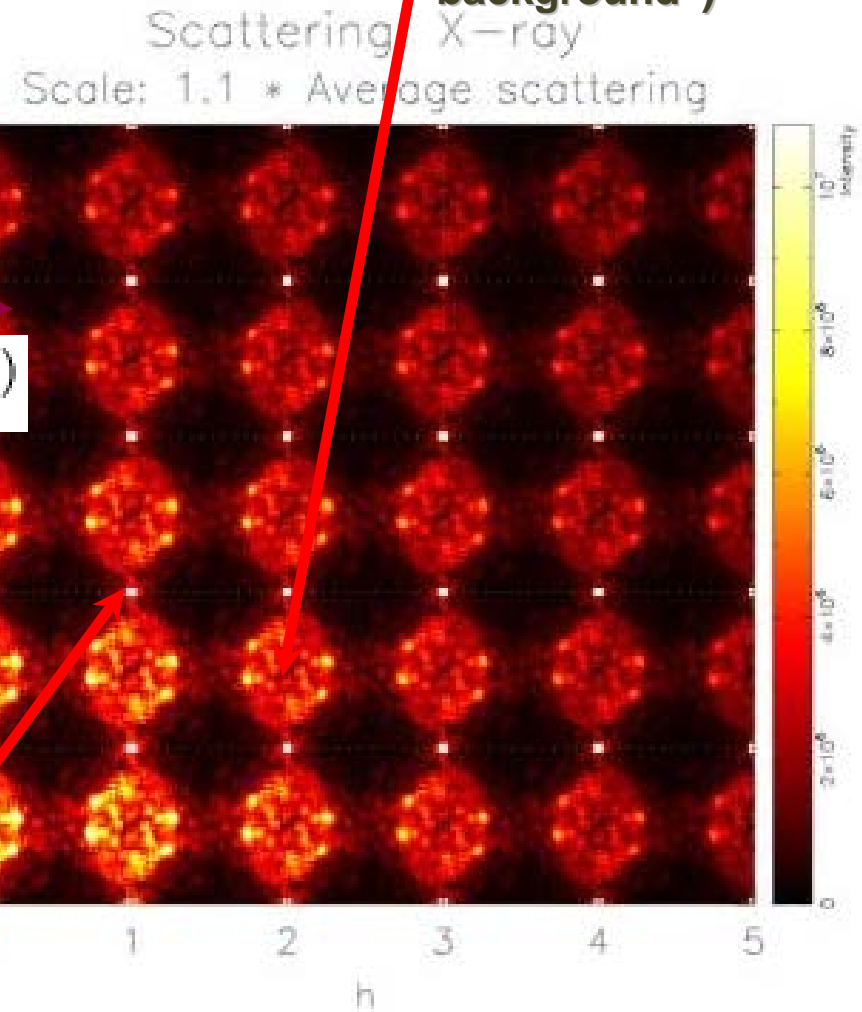
24% vacancies, 0.4 correlation in [100] (clustering along x)

Diffuse between and under Bragg peaks (usually treated as "background")



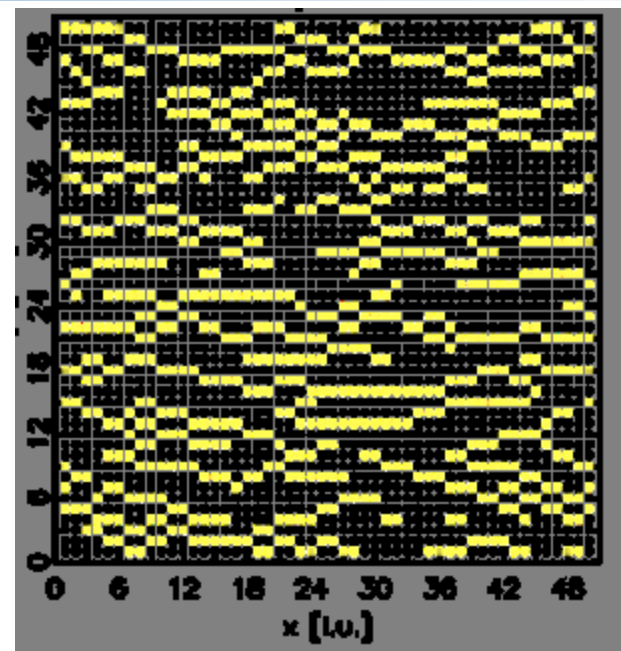
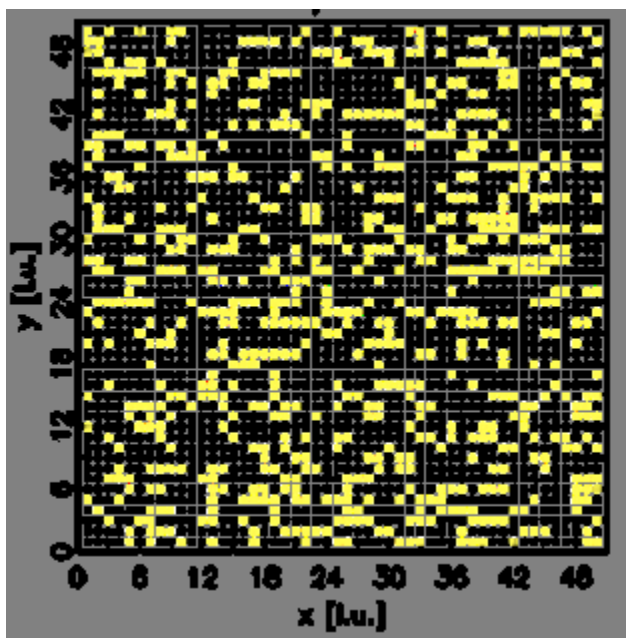
Bragg scattering

$$\mathcal{F}_x[f(x)](k)$$

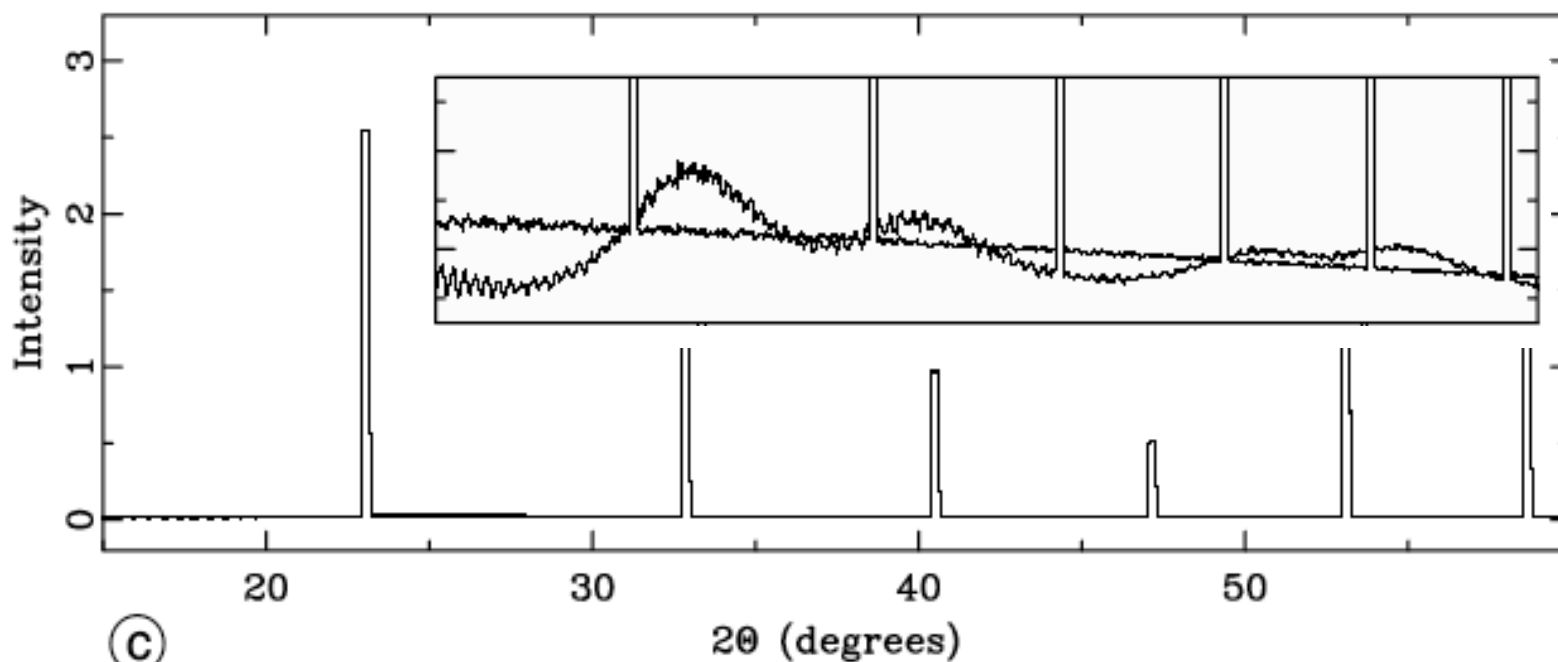


<http://www.totalscattering.org/teaching/>

“Random” and correlated vacancies



observation of Bragg scattering alone does not distinguish these models: lie atop one another. Structured “background” = diffuse scattering from correlation



Loss of information

Increase in diffuse scattering (deviations from “average” structure) merging of Bragg peaks

Loss of Rietveld-like refinement strategies

Too many parameters to define the scattering system

Too few data (peaks in reciprocal space are now broadened and merged)

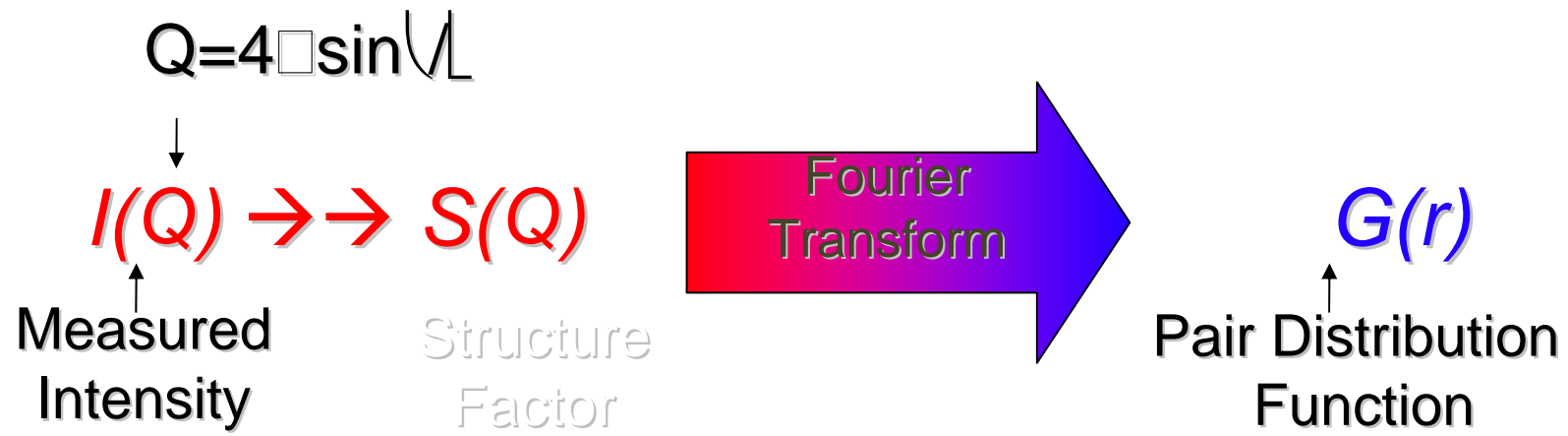
Some of the information is still there (just not as obvious)

Fourier transform TOTAL scattering (Diffuse + Bragg) and work in real space

Develop real space models (based on perfectly periodic models, guess, Monte Carlo....) and Fourier transform to produce to reproduce diffraction pattern

Obtaining the Pair Distribution Function for Powders:

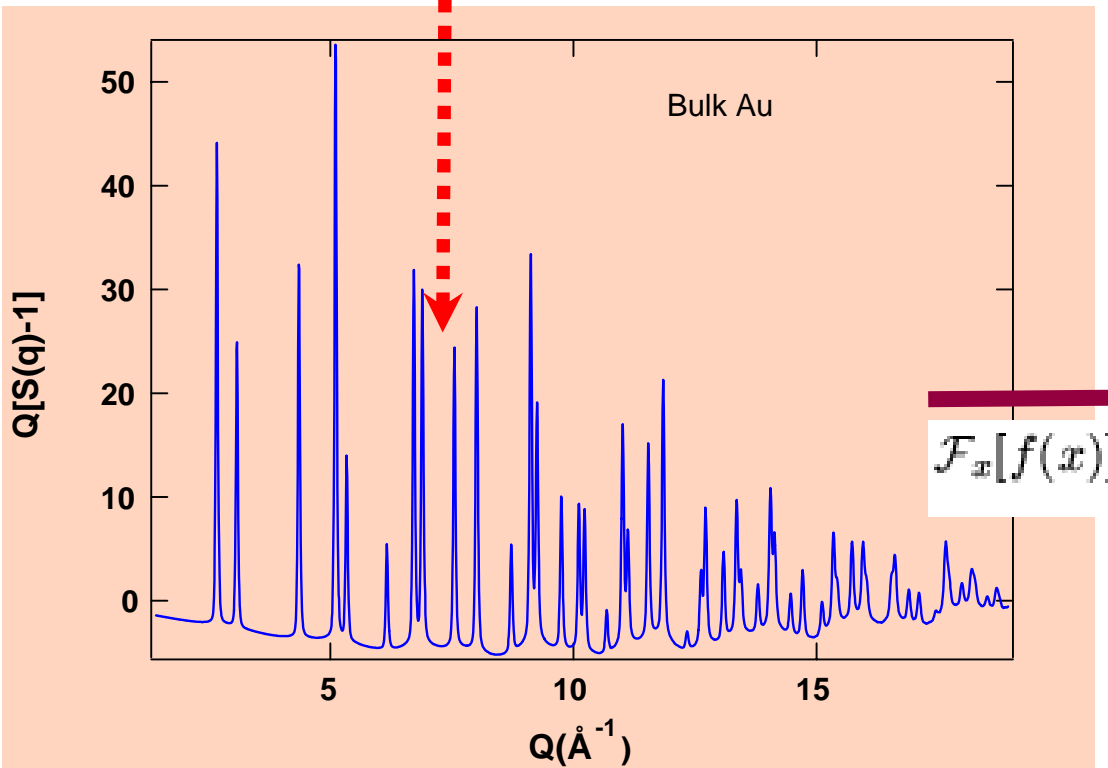
An Experimentally Accessible Quantity from FT of Total Scattering (Bragg + Diffuse)



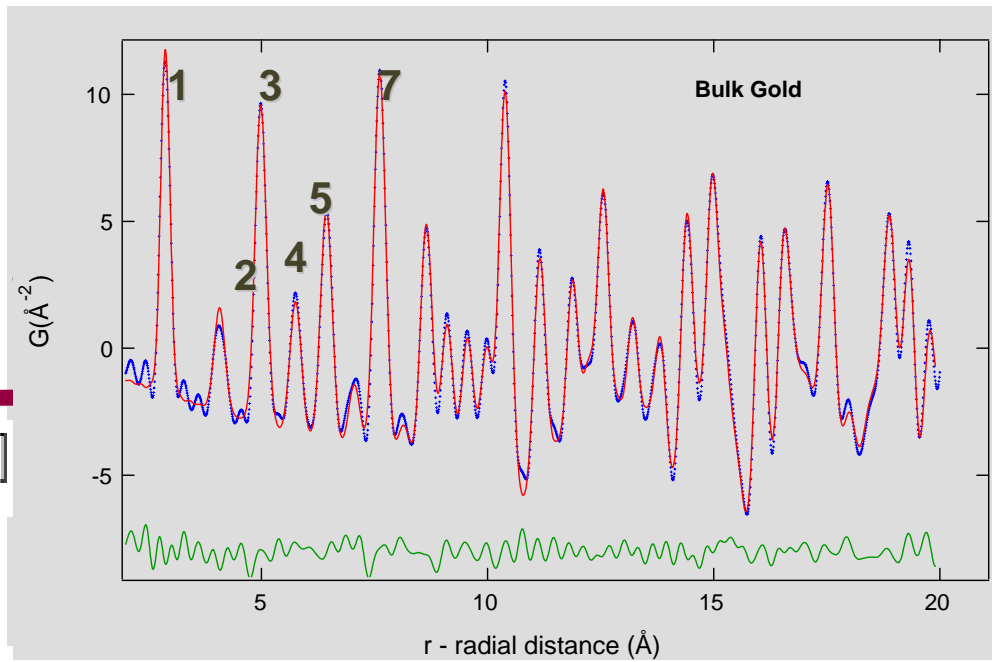
$$G(r) = (2/\pi) \int Q[S(Q) - 1] \sin(Qr) dQ$$

$$G(r) = 4\pi r [\rho(r) - \rho_0]$$

Limit of lab observations

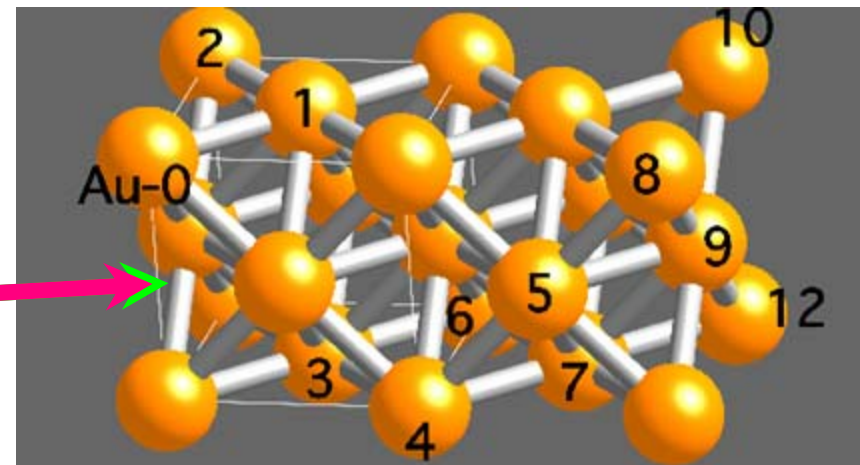


$(G(r) + \text{fit})$

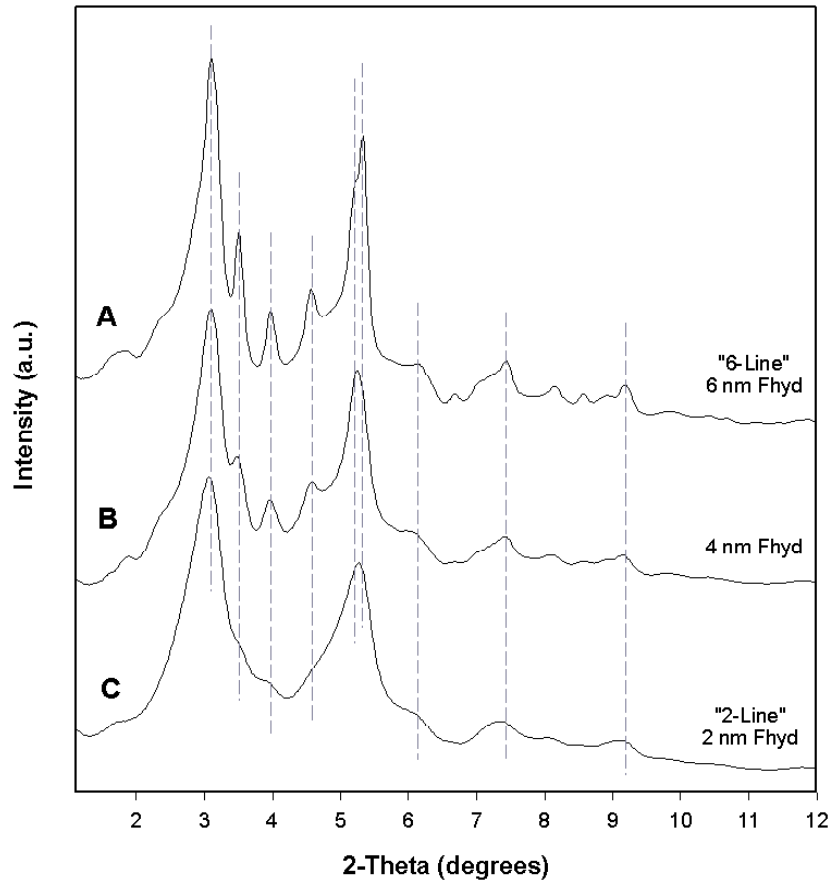


scattering $(S(Q) - \text{what we observe})$

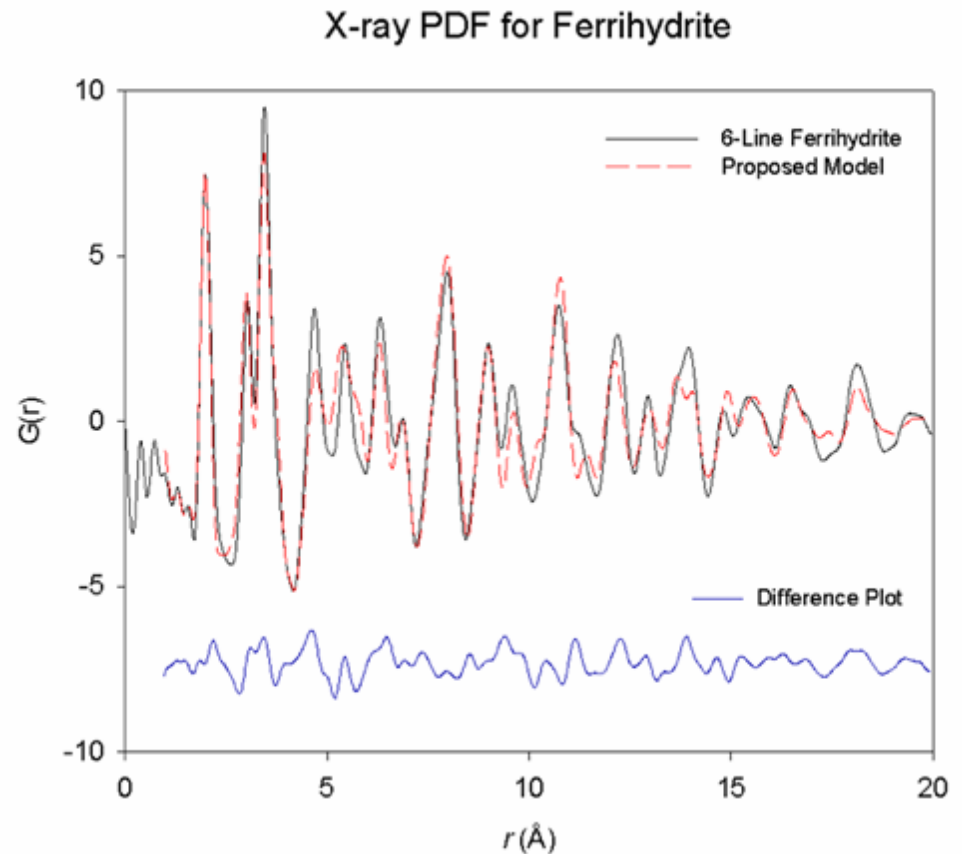
Adjust this model, calculate the red curve to minimize difference with “observed” $G(r)$ - the green curve



Comparing possible models - multiphase + disordered phase proposed for "structure" of ferrihydrite



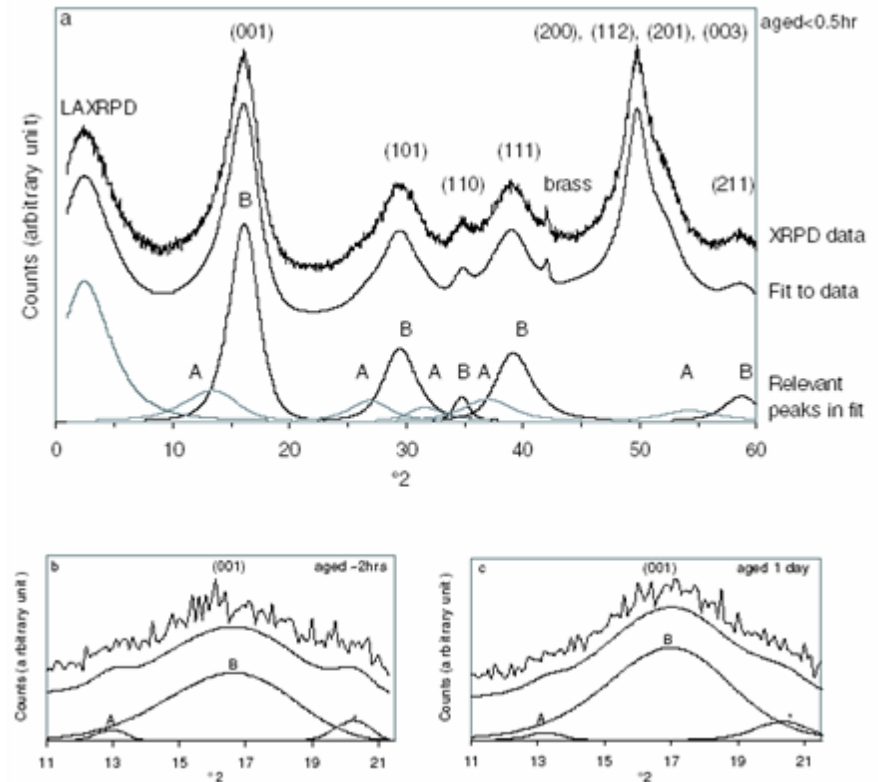
**** 2- and 6-line ferrihydrite (11-IDB, APS $\lambda = 0.13724 \text{ \AA}$) are they different? We obtain these patterns from a large number of preparation routes (ferritin, precipitation, aging...) Need to probe possible heterogeneities with nm - μm beams**



We obtain a credible fit with a single phase

Debunking models: is nano-crystalline FeS Single phase - or two phase?

- Research dates back to early 1900's
 - Berner, 1967
 - Rickard, 1975; 1997
 - Benning, 2000
 - Widler & Seward, 2002
 - Wolthers, 2003
 - Multi-phase structure models
 - Particle size ranging from 2 – 400 nm!
 - Estimated using various techniques:
 - BET
 - TEM/SEM
 - XRD (low-energy)
-
- **Opportunity to resolve controversy**
 - **Application of PDF technique**
 - **Fully characterize fundamental properties**
 - **Provide the foundation for more complex studies**



Wolthers, 2003

Geochemistry and environmental mineralogy
of the iron-sulphur-arsenic system

Two phases:

MkA = $2.2 \times 1.7\text{ nm}$

$a = b = 4.02\text{ \AA}$, $c = 6.60 \pm 0.1\text{ \AA}$

MkB = $7.4 \times 2.9\text{ nm}$

$a = b = 3.65\text{ \AA}$, $c = 5.48 \pm 0.2\text{ \AA}$

30 % MkA and 70% MkB

Structure of Mackinawite - FeS

FeS₄ tetrahedral coordination

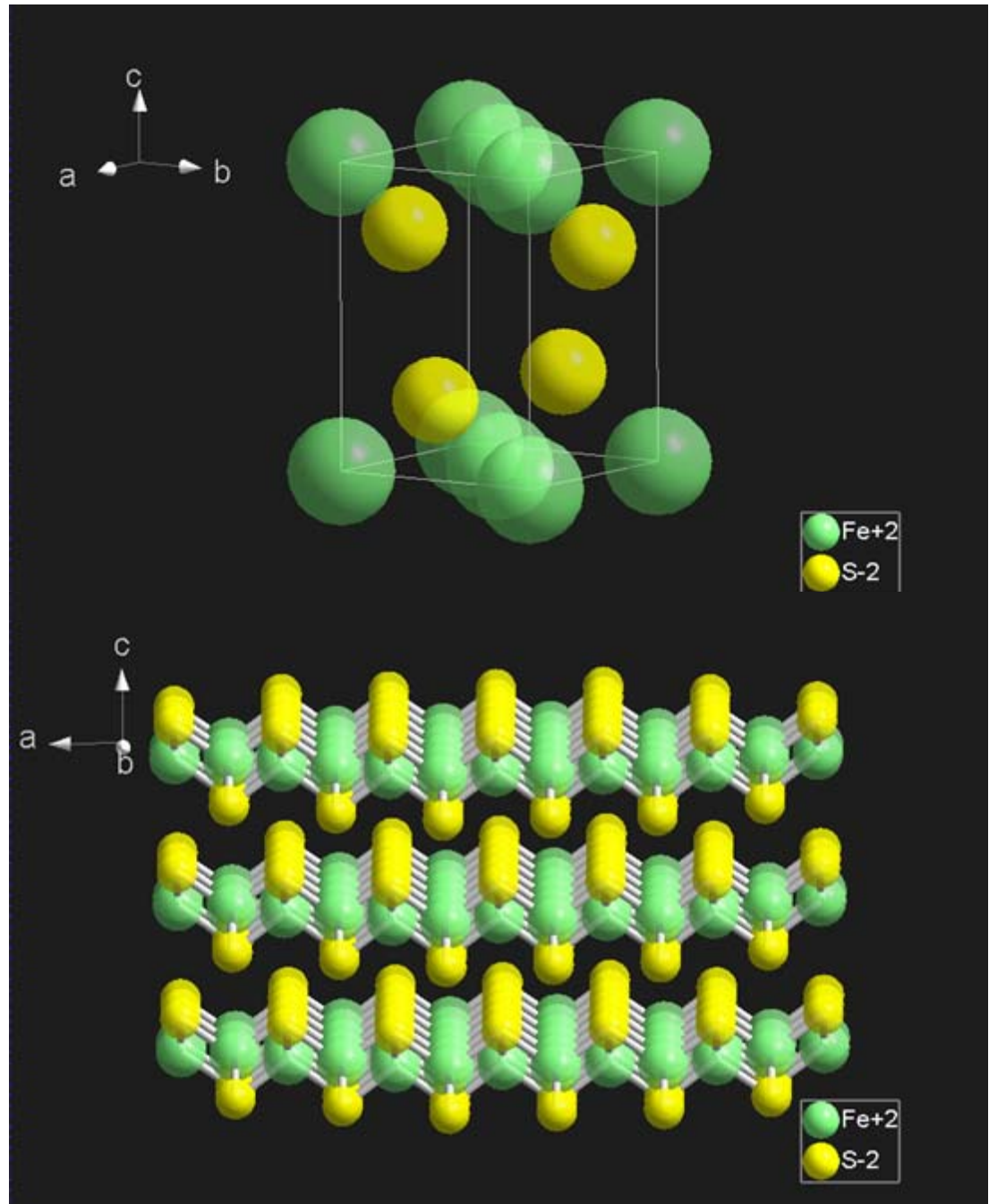
Crystalline FeS:

(P4/nmm)

$a = 3.676 \text{ \AA}$, $c = 5.032 \text{ \AA}$

Lennie et al., 1995

Highly susceptible to oxidation



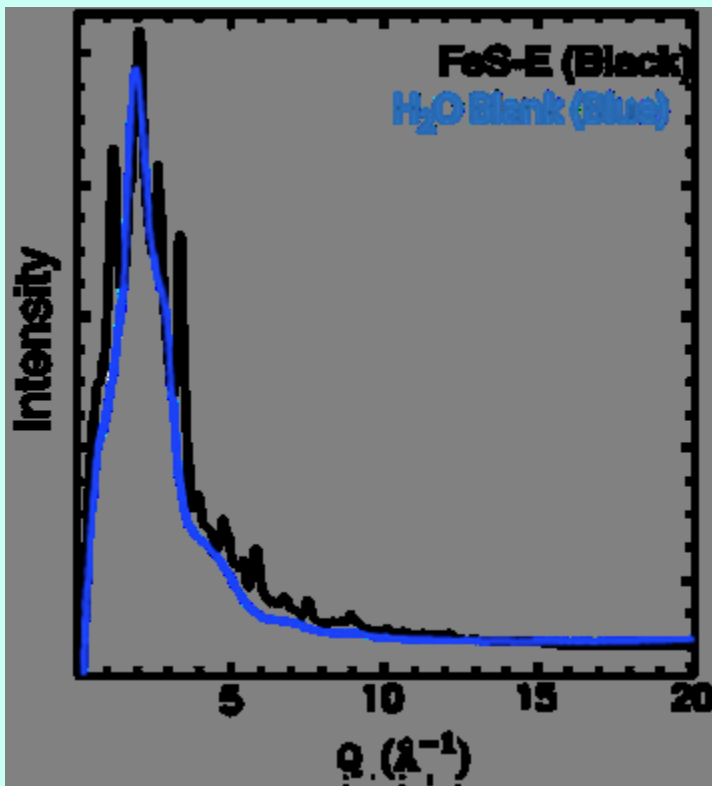
PDF: Normalization of $S(Q)$ - Total Scattering Structure Function

Normalization

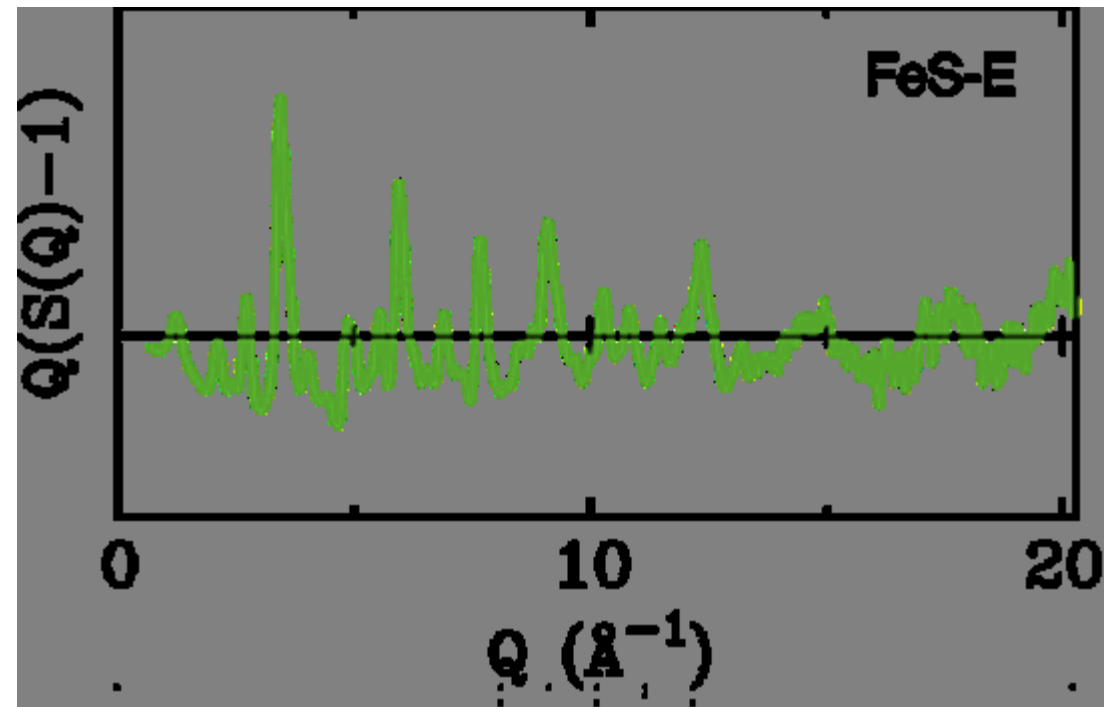
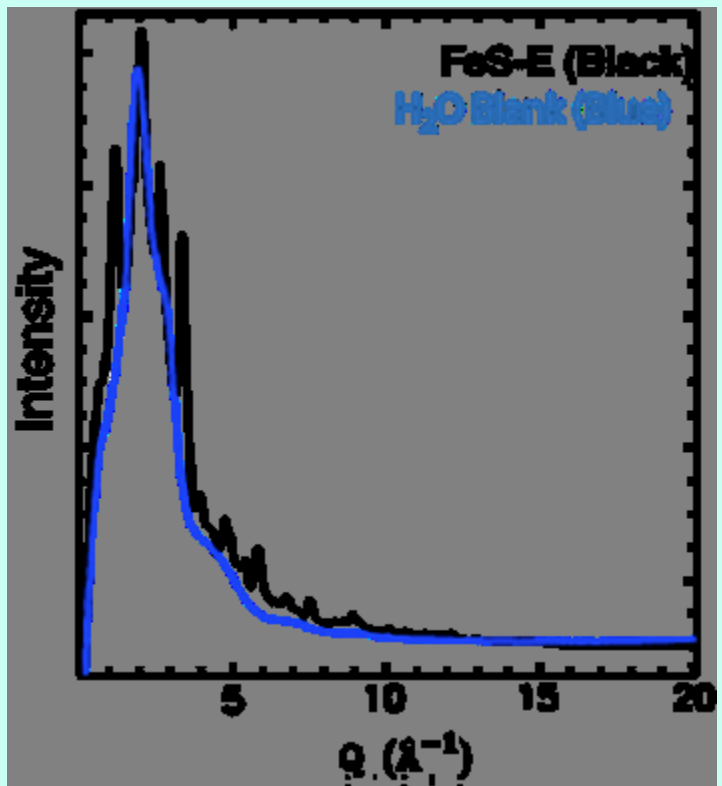
Use background spectrum (sample holder, medium, hutch...)

Scaling of background

Contribution of Bragg and diffuse scatter from SAMPLE ONLY



PDF: Normalization of $S(Q)$ - Total Scattering Structure Function



PDF: Test Modeling G(r) – Wolthers, 2003

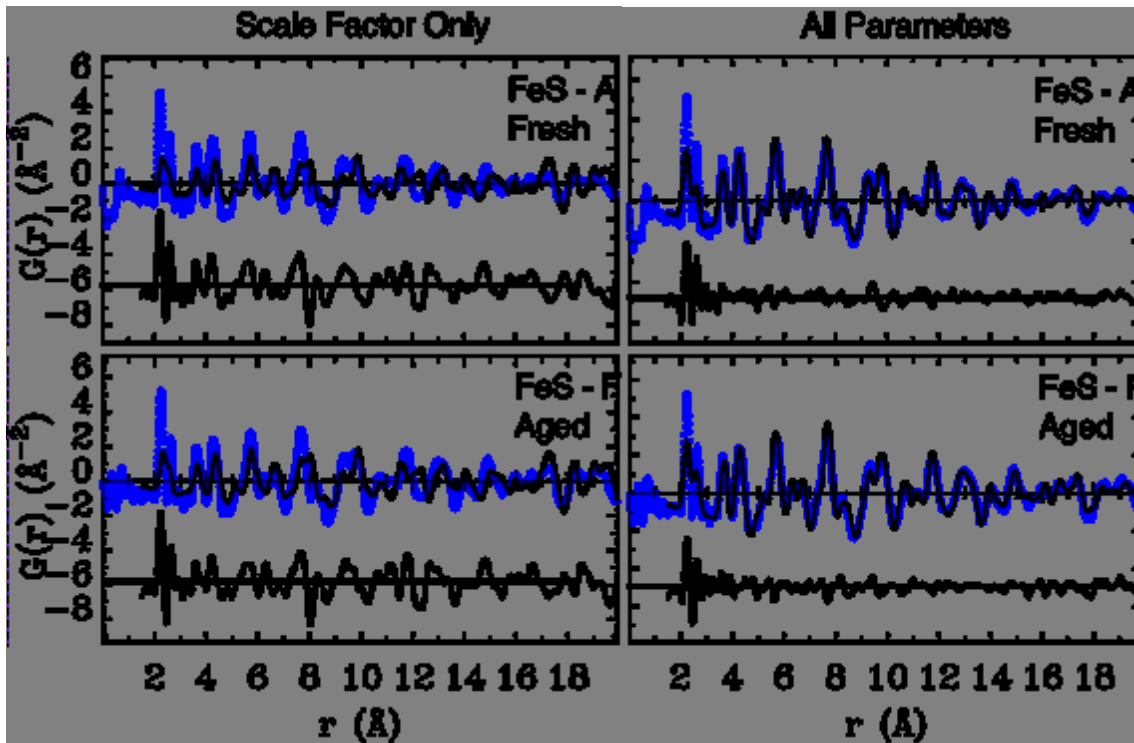
- Two phase:
 - MkA: Mackinawite
[a = 4Å, c = 6.6Å]
 - MkB: Mackinawite
[a = 3.65Å, c = 5.5Å]

- Phase Mixture:
 - 30% MkA
 - 70% MkB

- Aging Effects:
 - Decrease % MkA

*** Two-phase model does not fit**

*



FeS – A (Fresh)	MkA	MkB
Scale Factor	0 %	60 %
a – parameter (Å)	--	3.67
c – parameter (Å)	--	5.11
Rw	77.7% / 35.4%	
FeS – F (Aged)	MkA	MkB
Scale Factor	0 %	61 %
a – parameter (Å)	--	3.69
c – parameter (Å)	--	5.07
Rw	72.7% / 32.4%	
Rw	72.7%	

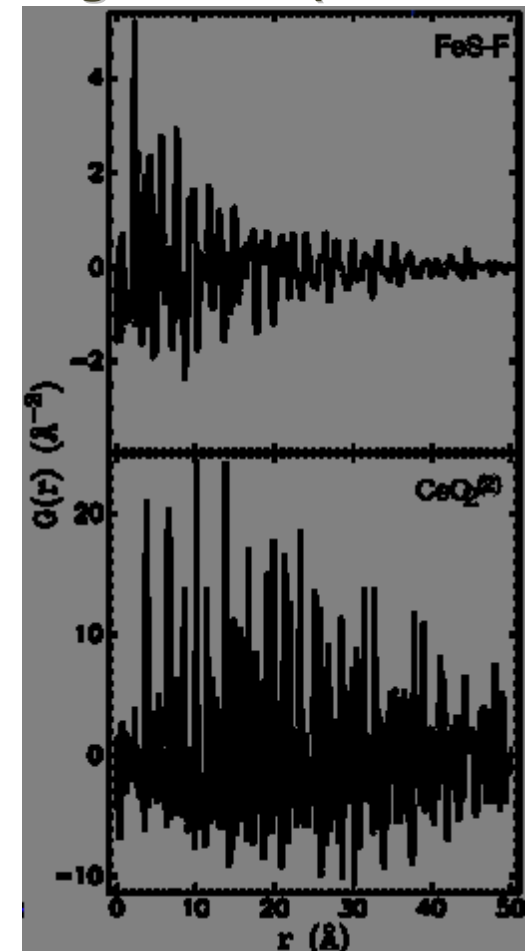
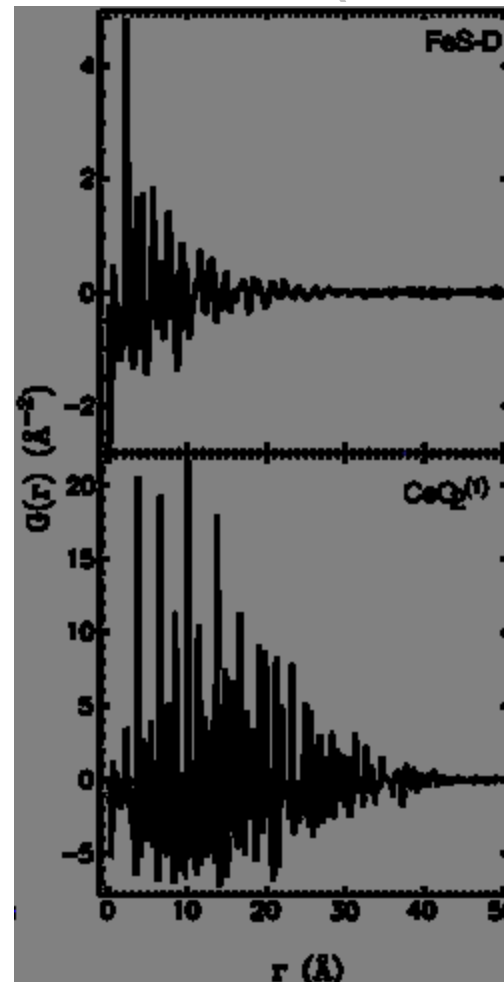
PDF: Range of Structural Coherence

- Fundamental particle size Range of structural coherence
Volume-weighted average maximum dimension of individual particles

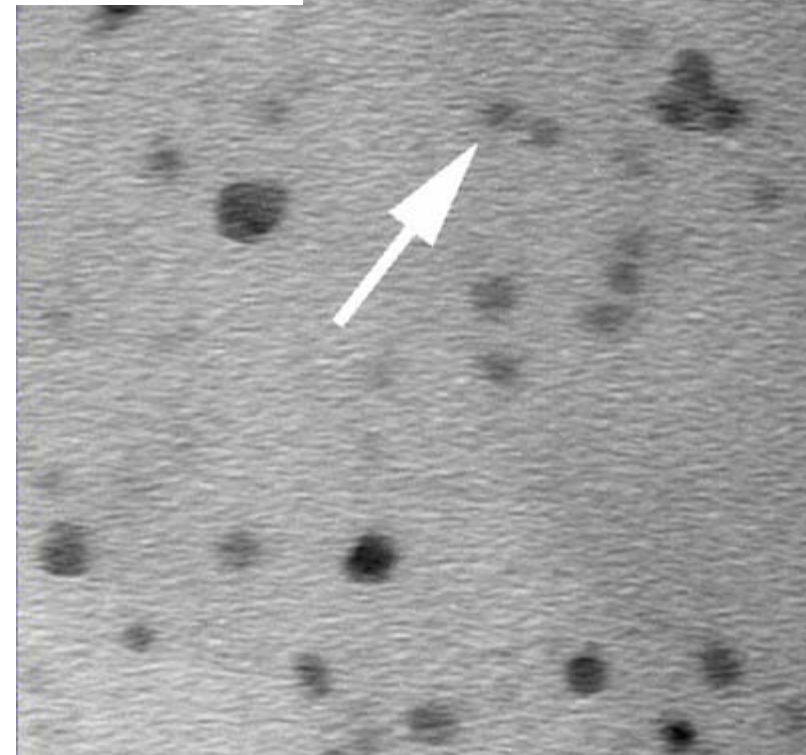
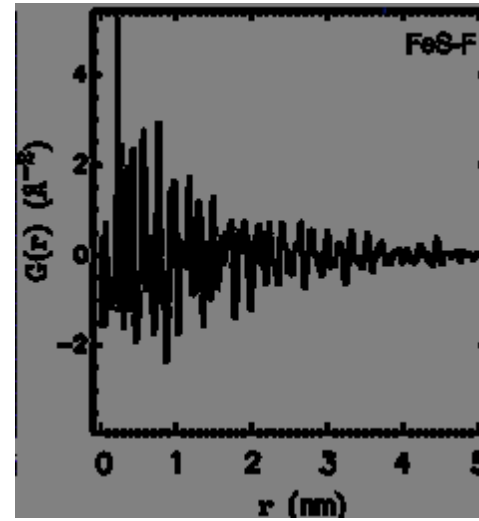
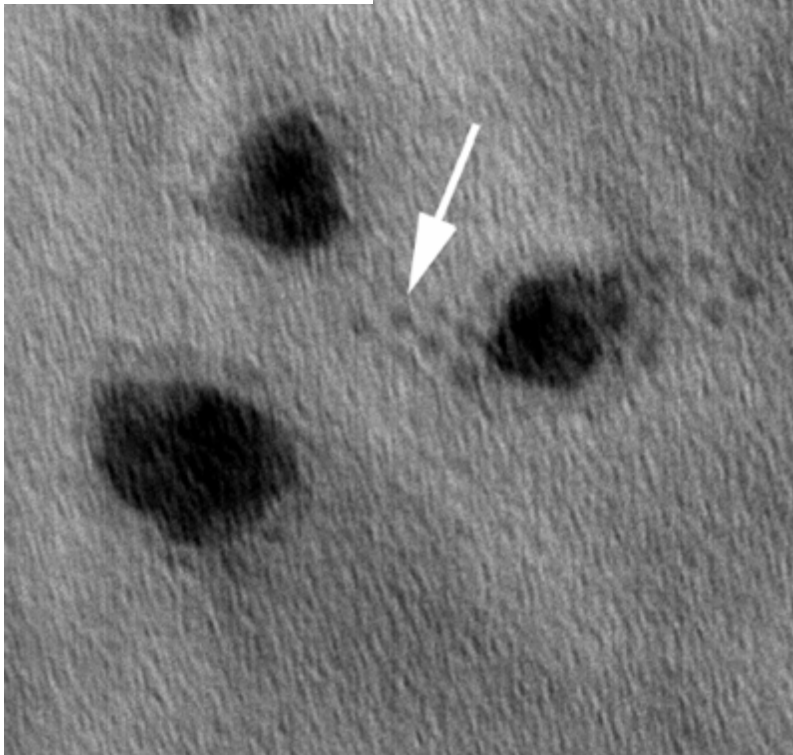
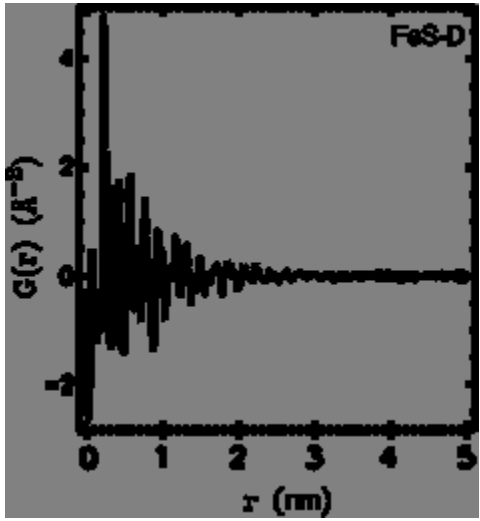
Assumes:
Single phase
Monodisperse
Within resolution
envelope of instrument

Fundamental particle sizes:
Fresh FeS ~ 2 nm
Aged FeS ~ 4- 4.5 nm

Fresh PPT (11-IDC) Aged PPT (1wk 70°C)

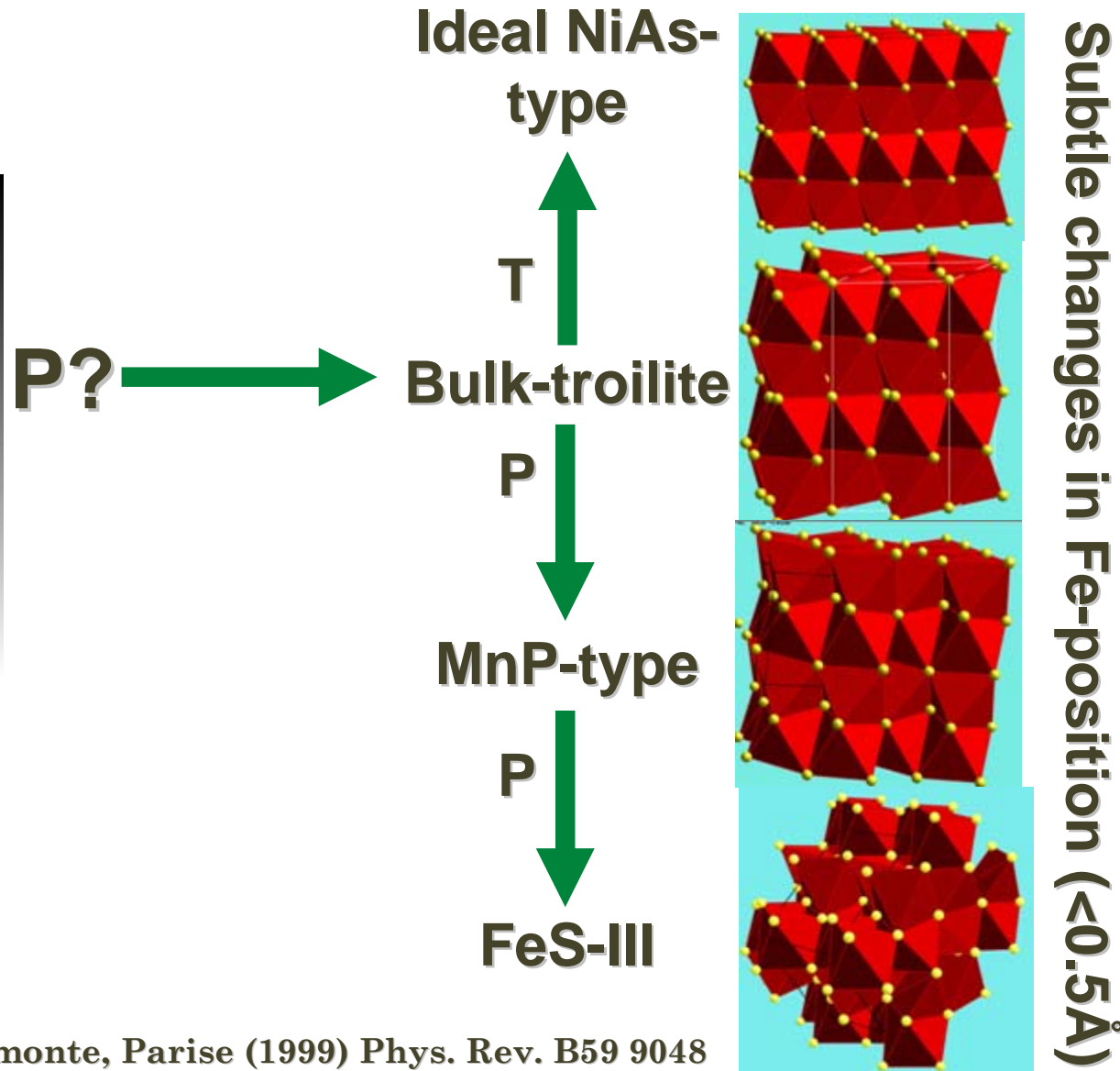
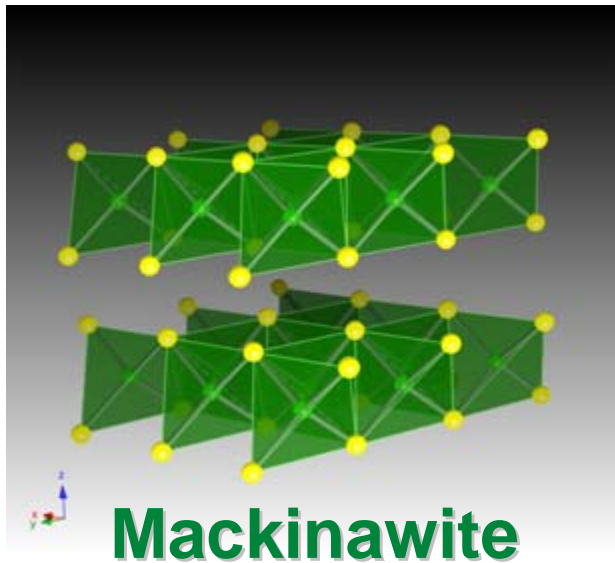


TEM: Fresh vs. Aged FeS



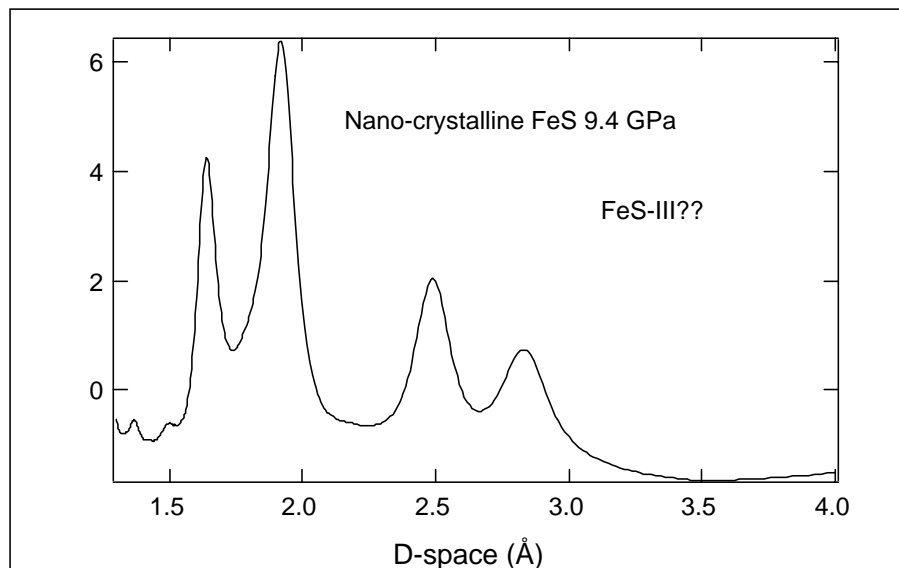
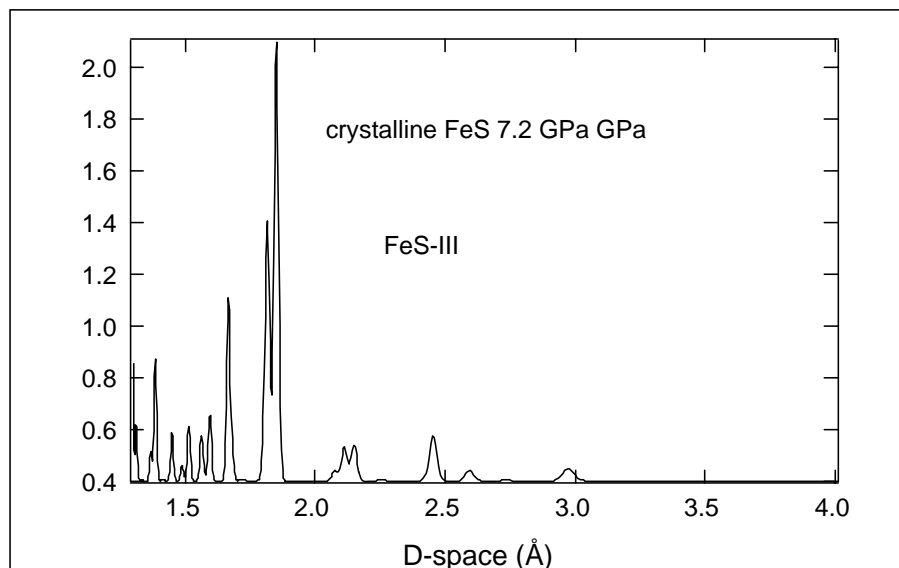
nano-crystalline FeS (mackinawite)

Same transformation pathways at high P as bulk FeS (troilite)?



Nelmes, McMahon, Belmonte, Parise (1999) Phys. Rev. B59 9048

FeS-III, stable above 7 Gpa - structure refinement from these data was challenging (~24 structural parameters) Nelmes et al PRB (1999)



Is this FeS-III? How do we expect to test this?

Energy

max Thompson ($E < 30$ keV),

min Chompton - practically independent of energy

Q-space resolution

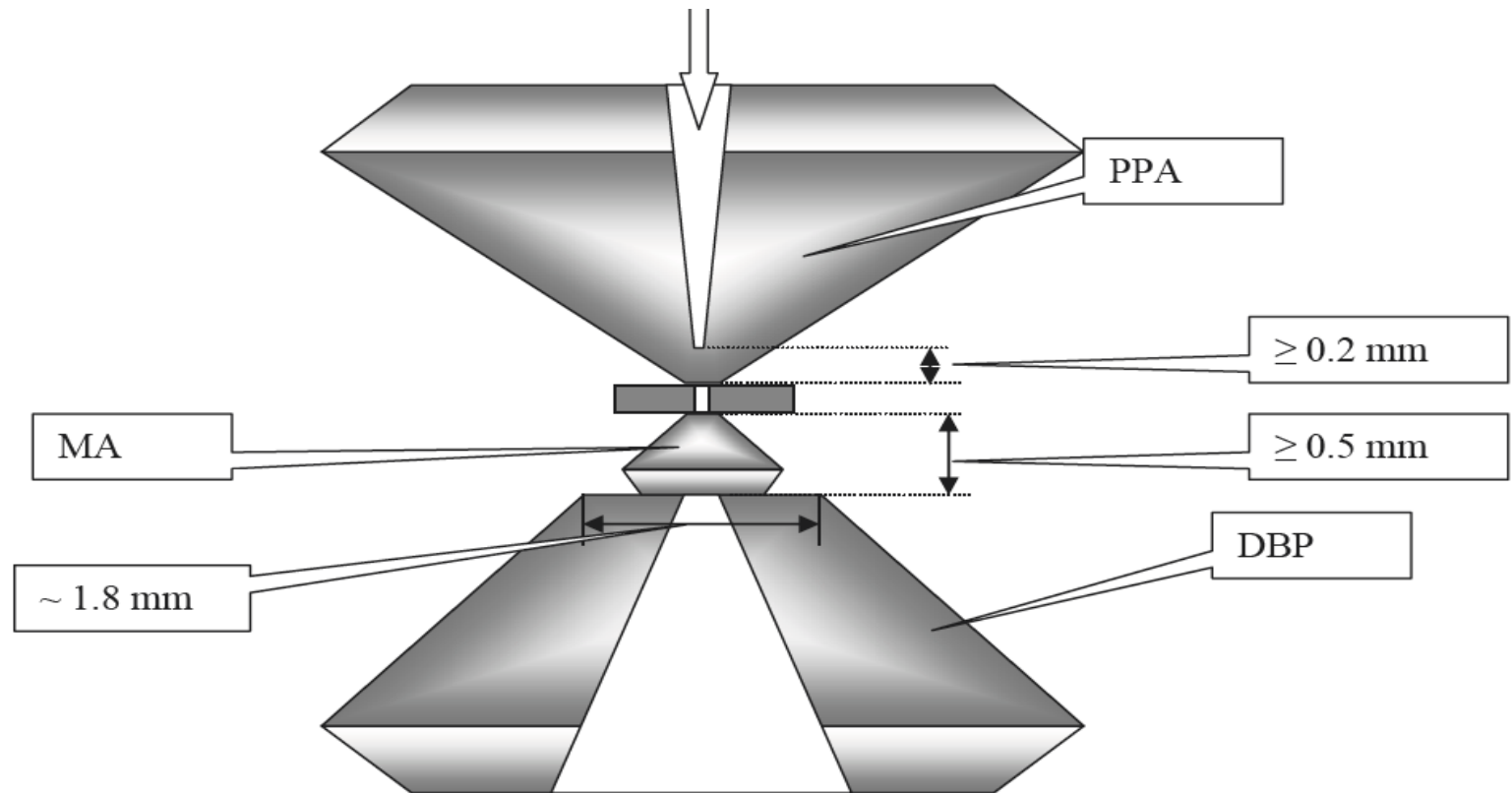
choose E, IP-sample distance for $Q_{\max} > 15 \text{ \AA}^{-1}$

Interference from DAC

heavy scatterers (Au, Ag) not a problem (see Martin et al J Appl Phys, 2005; Parise et al., J Synch Rad. 2005)

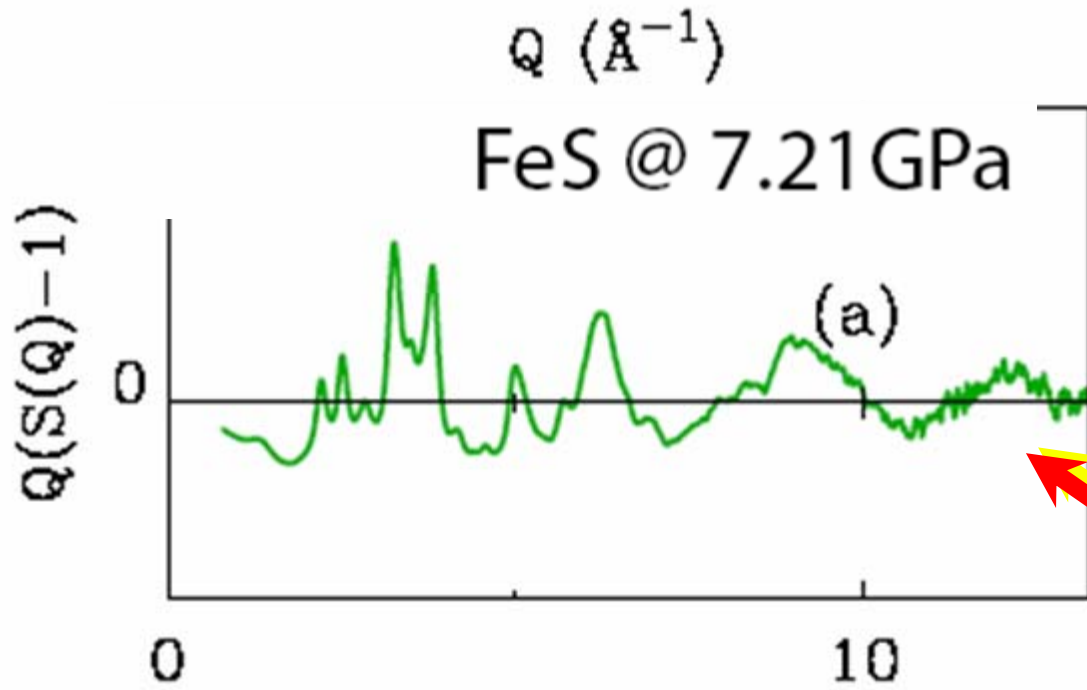
light scatterers - minimize diamond in beam
perforated diamond

Use large sample volume

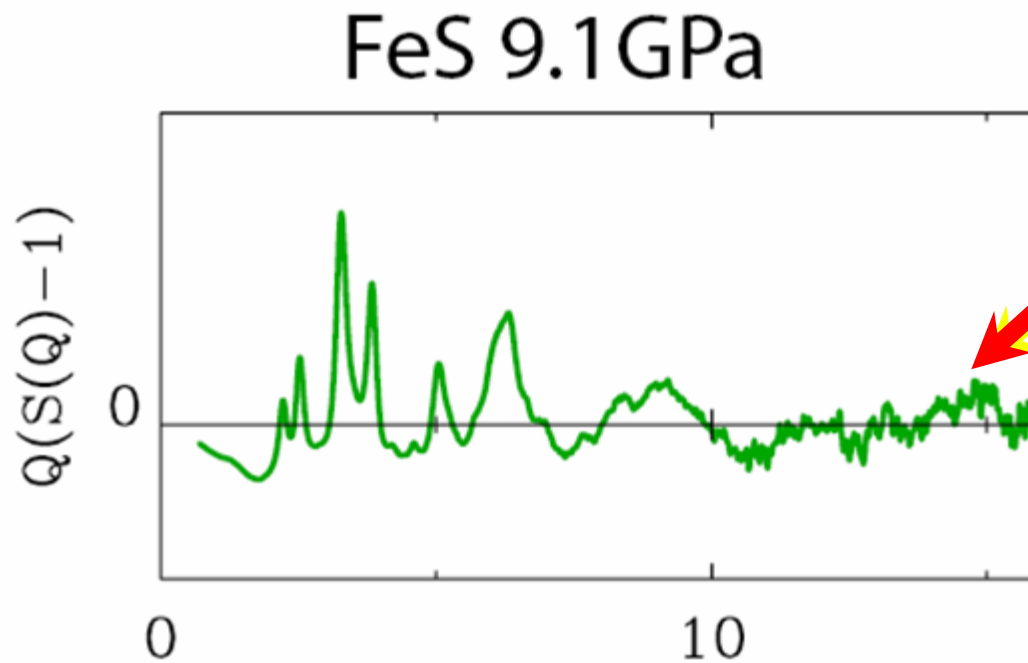


Why 1-ID?

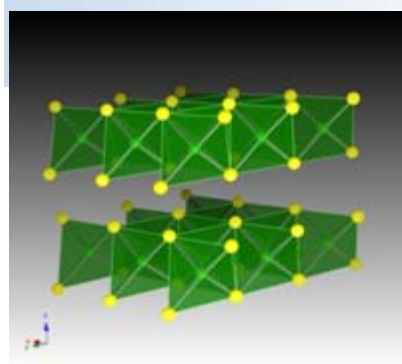
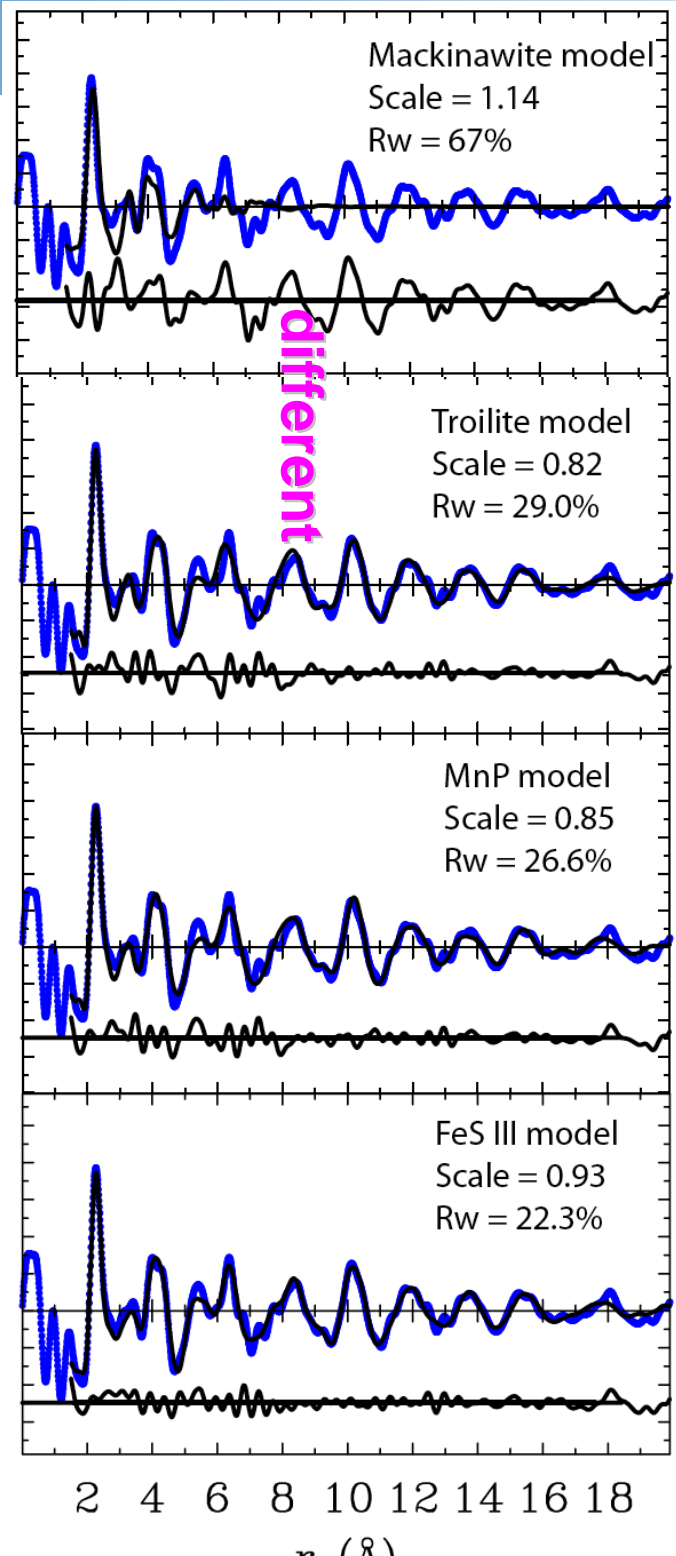
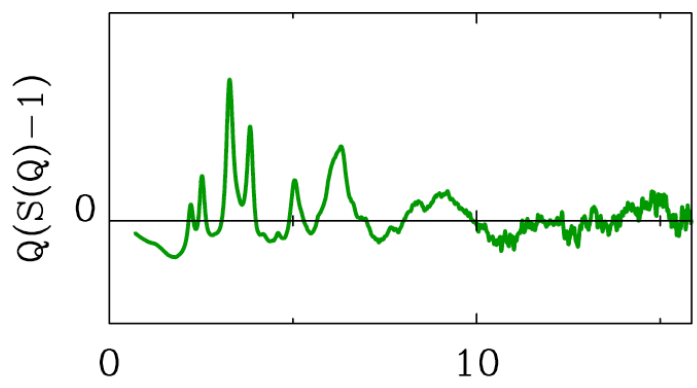
The sample is small, we need high energy, we need to get a beam down an 80 μ m hole



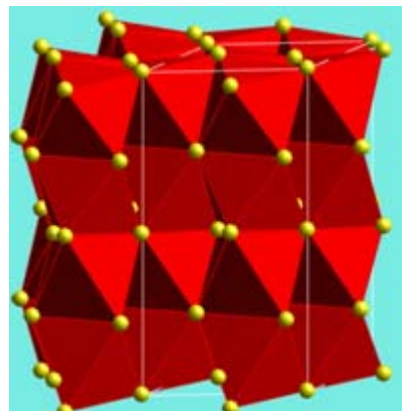
**there is not enough
signal out here**



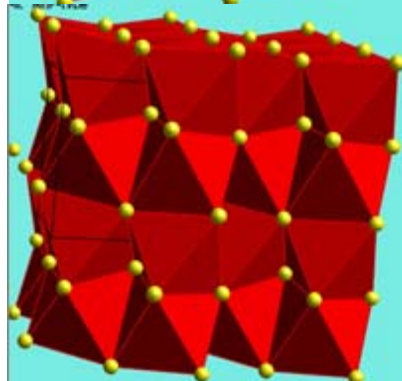
FeS 9.1GPa



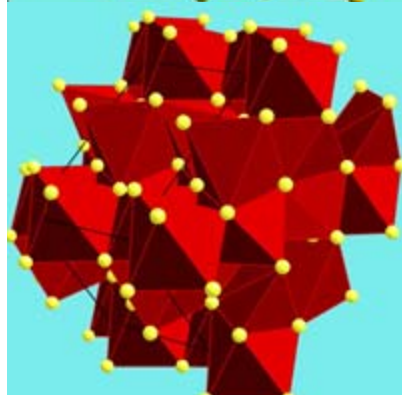
Mackinawite



troilite



MnP-type FeS



FeS-III

Michel, Antao, Martin, Chupas,
Lee, Schoonen, Parise April
2005

We know FeS! (no surface terminal groups....

**Moving onto “real” systems: need information *in situ*
scattering (still) a unique tool for in situ studies**

Relatively SENSITIVE when
large structural changes

Relatively INSENSITIVE

Very Subtle structural changes

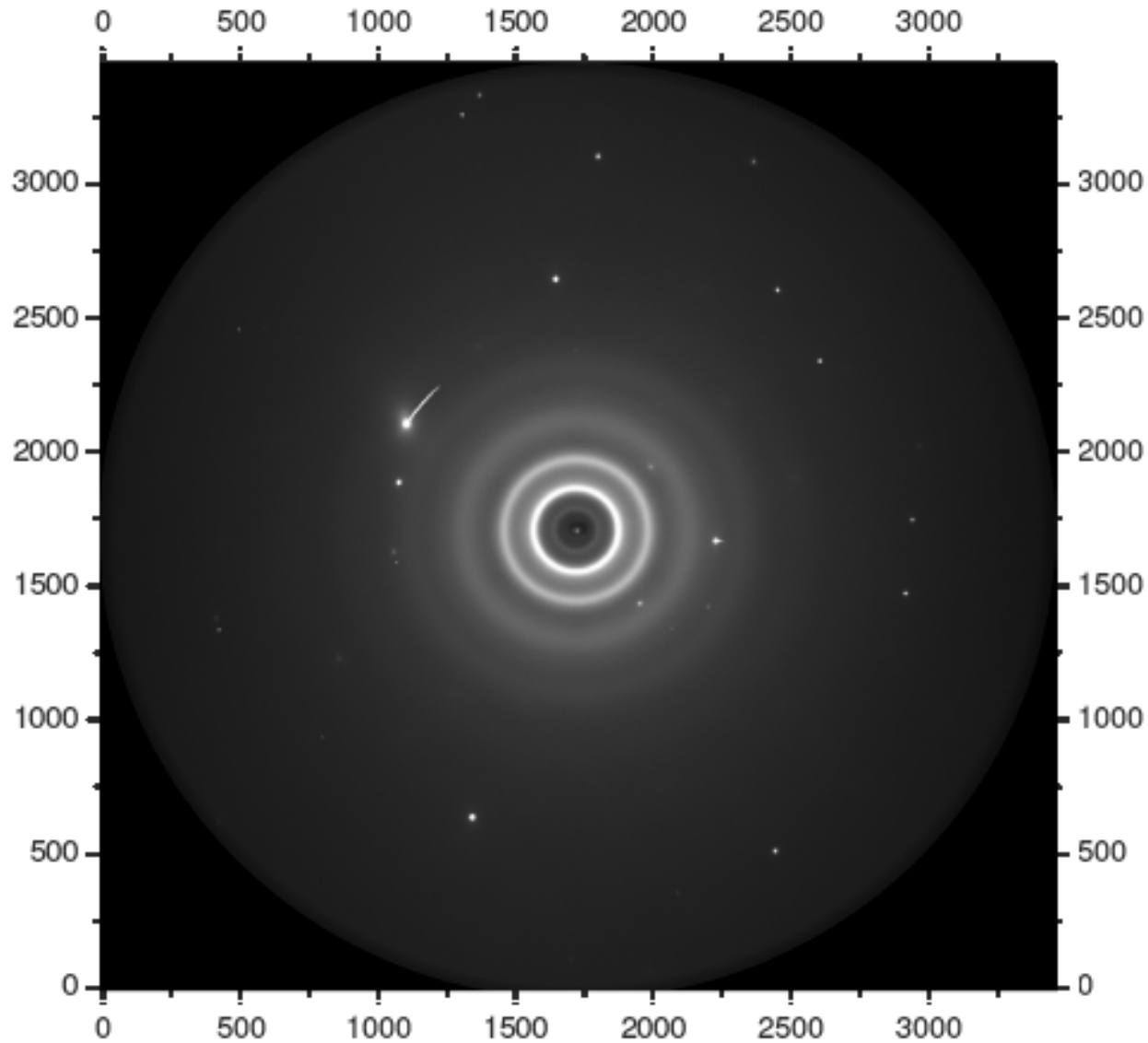
Mixtures of closely related phases

Solution: differential PDF?

Neutrons: use +v/-ve isotope (Bréger, Grey, Parise JACS 2005)

X-rays: anomalous scattering

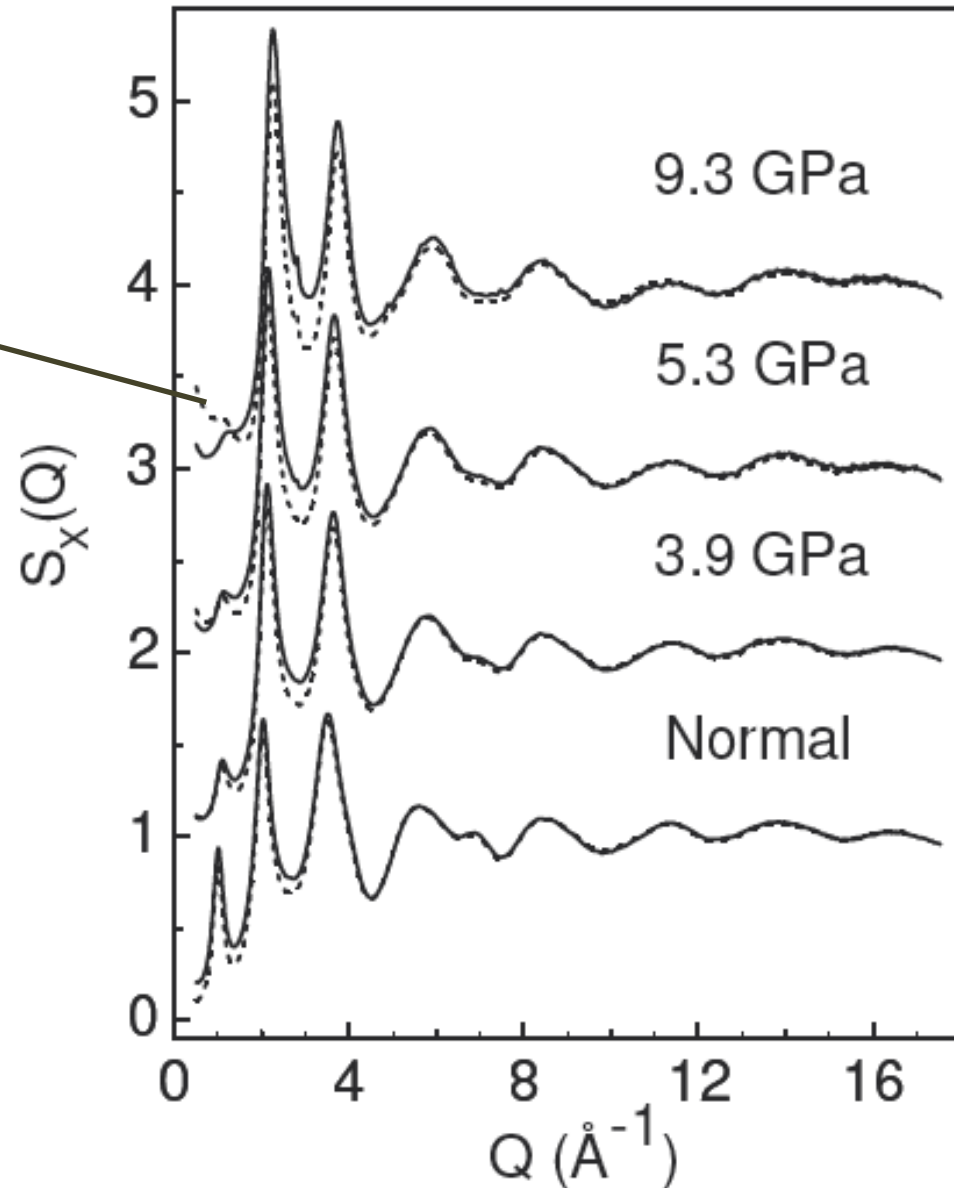
**New Monte Carlo tools combining techniques (NMR,
scattering.....)**



Incident x-rays of energy 80.047(3) keV. Bright spots are Bragg peaks from the single crystal diamonds - diffuse rings from GeSe₂ glass.

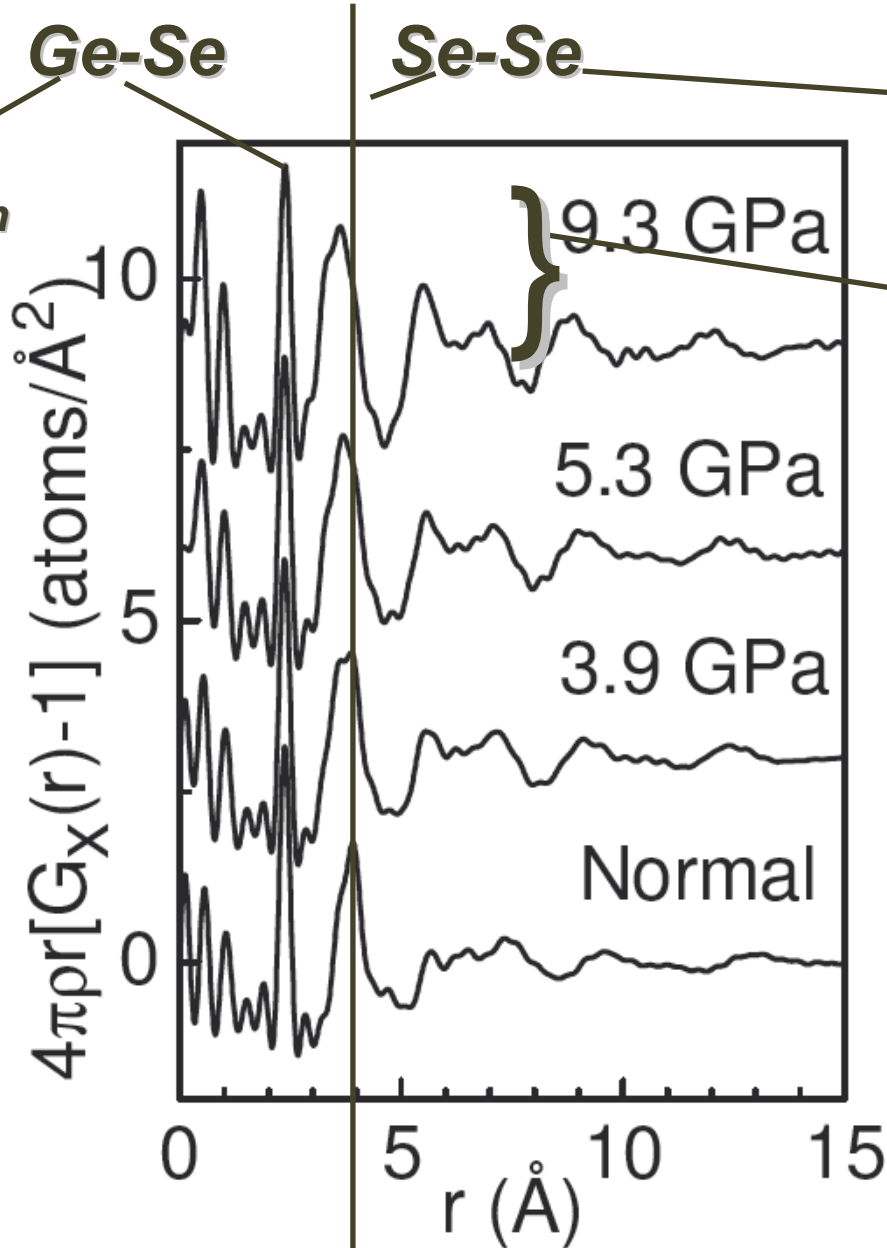
High pressure data from GeSe₂ - Q- space

Decrease in FSDF indicative of breakdown of intermediate range order



High pressure data from GeSe₂ - real space

Ge-Se distance increases slightly: 2.364(5) to 2.376(5)Å; GeSe_x coordination number increases x = 4.0(2) to x= 4.5(2) at 9.3 GPA



results consistent with distortion and breakdown of the tetrahedra during compression.

Peaks shift to low D, increases in intensity with increasing pressure, Consistent with increase in intermediate range order although the FSDP is significantly reduced

Intermediate range order

Complicated by overlap of distances

Need correlations from a number of techniques

Neutron scattering and isotopic substitution can help

Anomalous scattering at Ge and Se edges

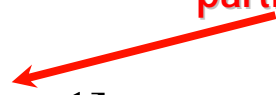
X-rays can not do somethings - we need to look at neutrons and Raman as well

Theory

The total structure for neutrons or X-rays is given by:

$$S_N(Q) = \frac{1}{\langle b \rangle^2} \sum_{\alpha, \beta} c_\alpha c_\beta b_\alpha b_\beta [S_{\alpha\beta}(Q) - 1]$$

partial structure factors



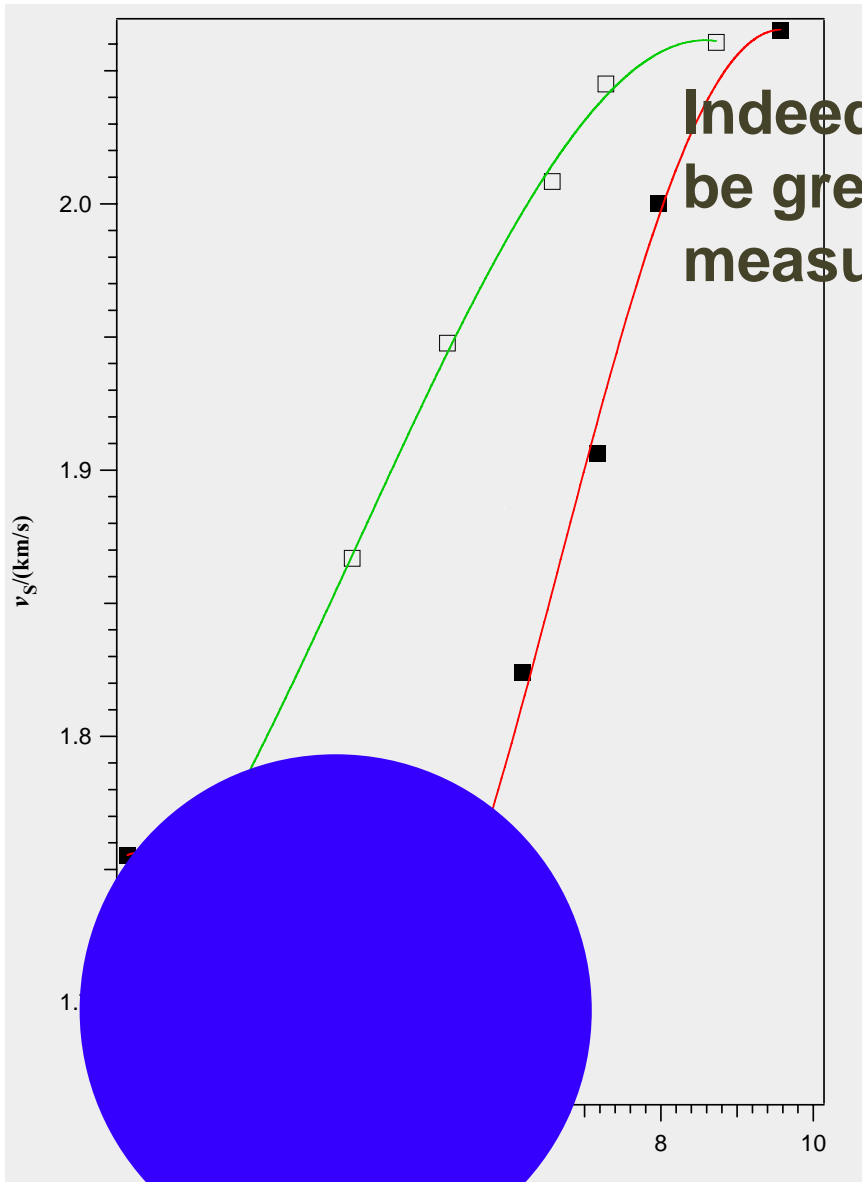
where c and b are the atomic fraction and coherent neutron scattering length (X-ray form factor). For BOTH X-ray and neutrons, the weighting factor $S(Q)$ are very similar $Z_{\text{Ge}}=32$, $Z_{\text{Se}}=34$; $b_{\text{Ge}}=8.19$ fm, $b_{\text{Se}}=7.97$. The pair distribution function is related to the Fourier transform of the total structure factors $S_N(Q)$ and $S_X(Q)$.

Because Z and b for Ge and Se are so similar the pair correlations

$$S_N(Q) \sim S_X(Q) \sim 0.115S_{\text{GeGe}}(Q) + 0.437S_{\text{SeSe}}(Q) + 0.448S_{\text{GeSe}}(Q)$$

solutions?: Anomalous X-ray scattering (12 keV); isotopic substitution for neutron - or choose another glass (perhaps wisest choice!)

These changes in intermediate range order reflected in property changes?



Indeed they are - and suggests there should be greater integration of property/structural measurements

Shear wave velocity as a function of pressure for GeSe_2 glass at RT on compression (red) and decompression (green)

Antao, Li et al (2006)

Useful to develop structure models for perturbed structures

Good for debunking models. *A priori studies?*

The nano- and glassy side is still developing

Quantitative

Properly normalized $S(Q)$ - range of $Q \gg 15\text{\AA}^{-1}$

For moderate pressures (< 10 GPa) WE have found the large volume device (Panoramic/GEM cells, PE cells) give superior data, even with very weak scatterers

The ERL

See about half of what was discussed in the dynamic group